

Structural, Electrical and Magnetic Properties of Zn Doped NiFe_2O_4 Added with 0.8wt% V_2O_5 Synthesized via Ceramic Method

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Abstract. Using the traditional ceramic process, zinc doped ‘Nickel ferrite’ with V_2O_5 was produced, yielding $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4 + 0.8\text{wt}\%\text{V}_2\text{O}_5$. Various technical methodologies were used to characterize structural, electrical, magnetic, and dielectric properties. X-ray diffraction was used to identify the crystalline phases after the addition of V_2O_5 as an impurity. Major reflections obtained in XRD were indexed as (220), (310), (311), (222), (400), (422), (511), and (440) planes that confirmed the formation of ferrite crystalline phases in particular with single-phase, cubic shape, spinel structure. The avg. lattice parameter (a) of the prepared sample was found to be 8.380 ± 0.005 Å. The X-ray density was recorded to be 5.303 g/cm^3 . The crystallite size (t) was calculated by using the Debye-Scherrer method taking the (311) plane of maximum intensity; it was reported to be $2.684 \mu\text{m}$. The development of the ferrite phase was supported by different absorption bands ν_1 (542 cm^{-1}) and ν_2 (410.84 cm^{-1}) corresponding to the tetrahedral and octahedral group complexes $\text{Fe}^{3+}-\text{O}^{2-}$ respectively. The logarithmic plot of DC resistivity has shown the semiconducting behaviour of the prepared samples. The magnetic parameters have shown some changes at higher levels after the addition of V_2O_5 in $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4$.

KEY WORDS: Nickel ferrite, ceramic method, V_2O_5 , X-diffraction, magnetic property.

1 Introduction

Magnetic materials are currently popular due to their uncompensated physico-chemical properties, which make them stand out in a variety of applications. Following the challenges in the scientific and technological domains, the global demand for novel materials and spinel ferrites is expanding linearly. Due to

their possessive semiconducting activity, electro-magnetic nature, electrochemical capabilities, and other characteristics that they carry with them separately and in the sharing, these 'iron oxides' (ferrite) have various and unobjectionable requirements. [1]. The numerous applications that meet global demand in the fields of magnetic fluid technology, nano-electronic chips used in mobile/computers, gas sensor devices, microwave technology, transformer core, transducer, nano SMPS, motor, and dynamo [2–4]. Spinel ferrite is used in significant diagnostic equipment in the medical field. In addition, ferrite has accepted new challenges in the construction of circulators, power equipment, memory devices, and other electronic components. Ferrites perform admirably as a performance enhancer in current energy storage devices such as electrodes of electrochemical cells, exchangers, transporters, and so on. Magnetic spinel ferrite has overcome biocompatibility issues in biotechnology and medical science [5]. They worked efficiently in biosensing devices, isolators, phase shifters, MRI, spintronics, etc. [6, 7]. Ferrites' spontaneous physical and chemical qualities make them suitable for a wide range of applications, including pharmaceuticals, agriculture, industry, and home wastewater treatment. The spinel ferrites' synergistic impact is extended to the design and processing of novel materials. The progress of spinel ferrites in electromagnetic interference shielding is well documented. Spinel ferrite is also well-suited to innovative applications such as magnetic absorbents and photo-catalysts etc. [8, 9]. The spinel ferrites have a cubic lattice structure that can be expressed with a general formula AB_2O_4 [10]. Many academics from all over the world have struggled with the geometric engineering of the spinel's dynamic crystal structure. The crystal lattice's nomenclature belongs to the space group $Fd_{3m}-O_h^7$; involved tetrahedral (A)-site and octahedral (B)-site, in partnership with the Fe^{3+} ions [11]. The ferrite structure has eight tetrahedral gaps, which are inhabited by trivalent ferrous ions (Fe^{3+} ions), and sixteen octahedral voids occupied in combination by trivalent ferric ions (Fe^{3+} ions) and divalent metal ions viz. Ni^{2+} , Co^{2+} , Cu^{2+} , Mg^{2+} , Mn^{2+} , Ca^{2+} , Zn^{2+} etc. [12]. In this category, nickel ferrite ($NiFe_2O_4$) is a soft ferrite with, inverse spinel [13, 14]. Various synthesis methods have been designed for research and commercial purposes in order to improve the performance chart, achieve new results, and fulfil the application pyramid of these ferrites. Some of the special synthesis methods are sol-gel auto-combustion [15], electro-deposition method [16], co-precipitation method [17], micro-emulsion [18], auto-clave method [19], salt-melt technique [20], spray pyrolysis [21], solid-state reaction method [22], and so on. The term 'additive' refers to a material that is mixed with another substance. An additive is a chemical compound that is added to a target material to increase its physical or chemical qualities while preserving the material's basic properties. Fillers, plasticizers, lubricants, colourants, and other additives must be stable in production conditions, non-toxic, and stable during use while taking availability and cost into consideration. The adding of additives to materials improves bulkiness, mechanical characteristics, chemical properties, optical properties, and rheological

properties. To achieve a desired change in physical properties, the addition of such additives is limited to a modest amount in compared to the host substance. Spinel ferrite's physico-chemical characteristics can be tuned by adding various chemicals to it. Several oxides like niobium oxide (Nb₂O₅), vanadium pentoxide (V₂O₅), tungsten (IV) oxide (WO₂), Lithium carbonate (Li₂CO₃), Bismuth(III) oxide (Bi₂O₃), Cobalt oxide (Co₂O₃) etc. are used as additives [23]. The chemical element vanadium has the symbol V and the atomic number 23. It's a transition metal that's hard, silvery grey, ductile, and malleable. Vanadium is a metal that is rarely discovered in nature and is thought to be poisonous, medium-hard, and ductile. It's a steel-blue transition metal with electrical conductivity and thermal insulator characteristics at its core. Vanadium is rarely discovered in its natural state, but the creation of an oxide layer passivates the free metal against further oxidation once it is isolated chemically. In 1801 in Mexico, Andrés Manuel del Ro incorrectly identified the components of an idea by examining a new mineral called erythronium [24]. Niels Gabriel Sefstrom was persuaded to show the existence of a new element, which he termed vanadium, a few years later in 1830 [25]. Unlike gold, silver, or even nickel, vanadium is widely used in a variety of items and construction materials. Vanadium is an important component in its major application, which is steel strengthening. It can be found in buildings and bridges all around the world, as well as vanadium redox batteries. In Quebec, Canada, roughly 500 kilometres north of Montreal, the mining capital of Chicago, one of the world's largest vanadium layers lies barely beneath the surface. Vanadium pentoxide is an inorganic compound with a high degree of oxidation, making it an oxidizing agent. Vanadium has a bcc structure, which means it has an orthorhombic crystal structure. The alloys of V₂O₅ are utilized as surgical instruments and tools, steel additives, ferrovanadium, vehicle crankshafts, axels, and dental implants due to their non-magnetic nature [26, 27]. Many researchers are reporting the need of V₂O₅ in the materials designing and applications in the various fields. Currently, Xiang Zhang et.al has reported in his literature about the high rate capability along with the ultralong cycle life densification role of free-standing V₂O₅/VG cathodes in their literature [28]. M Samir Ullah et. al has discussed about the concentration effect of V₂O₅ on the structural and magnetic properties of Ni-Co-Zn [29]. Recently, B. Deka Boruah et al. has studied the V₂O₅ cathodes used in the light rechargeable lithium-ion batteries. Kezhuo Li et. al have mentioned in their report the V₂O₅ effect on the porous ceramics [30]. Tarik T. Issa and Samara J. Mohammad has shown V₂O₅ as a candidate for the compacted composites ceramics Yttrium oxide 80 wt.% [31]. The structural, electrical, and magnetic properties of Ni_{0.7}Zn_{0.3}Fe₂O₄ + 0.8 wt%V₂O₅ were studied after spinel ferrite was produced using the ceramic method.

2 Experimental

2.1 Synthesis of $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4 + 0.8 \text{ wt}\% \text{V}_2\text{O}_5$

For the initial starting materials, a high purity $\geq 99.9\%$ metal oxides like nickel oxide (NiO); zinc oxide (ZnO); ferric oxide (Fe_2O_3), and vanadium oxide (V_2O_5) were used for the ceramic synthesis. Nickel zinc spinel ferrite of chemical formula $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4 + 0.8 \text{ wt}\% \text{V}_2\text{O}_5$ was prepared via ceramic method using mortar pestle and milling technique. The essential requirement for making the initial mixture of the selected oxides of all the initial compositions was weighed according to their molar ratios by a weighting machine. The mixture was homogenized using the agate mortar and pestle and ground with the help of a grinder. An obtained $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4 + 0.8 \text{ wt}\% \text{V}_2\text{O}_5$ as kept for heating at the temperature of 200°C aiming to the remove of moisture. Then thoroughly mixed powder $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4 + 0.8 \text{ wt}\% \text{V}_2\text{O}_5$ again kept for calcination at 900°C for the period of 24 h. The gained powder was again grounded for 3 h for better crystallization and tuning the crystallite size. This powder was pressed into small pellets shaped in a disc with the dimension of $10 \times 3 \text{ mm}$ using a hydraulic press. Polyvinyl alcohol (PVA) was dropped in the prepared mixture to bind the molecular structure of the synthesized material, which can be then used for the pellet formation. The complete sintering process was carried out at a higher temperature at 1080°C held in the alumina crucible in the programmable furnace for 6 h. The final yield product $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4 + 0.8 \text{ wt}\% \text{V}_2\text{O}_5$ in a powder form was obtained.

2.2 Characterization techniques

The X-ray diffraction pattern of $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4 + 0.8 \text{ wt}\% \text{V}_2\text{O}_5$ was taken by using $\text{Cu-K}\alpha$ radiation ($\lambda = 1.54056 \text{ \AA}$) on (Bruker X-Advanced D8) X-ray diffractometer. The XRD parameters were characterized at room temperature operated at 20 mA current 40 kV voltage. The full X-ray diffraction pattern was produced through ceramic technique was recorded in the 2θ range of 20° to 80° at a scanning rate of 0.01 deg/min, and the primary Bragg's reflections were recorded for structural parameter analysis. The infrared spectroscopy was recorded by NICOLET-NEXU-870 spectrometer using KBr pellet at room temperature in the wavenumber range of $0 \text{ cm}^{-1} - 4000 \text{ cm}^{-1}$. The IR spectroscopy of the $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4 + 0.8 \text{ wt}\% \text{V}_2\text{O}_5$ sample was carried out for the determination of functional group analysis. The structural behaviour and phase development were reported from the IR spectra by examining the various stretching vibrations inside the lattice structure. The pulsed-field hysteresis loop tracer (PFHL) supplied by MAGNETA Pvt. Ltd. Mumbai was used to measure the magnetic properties of the samples at room temperature. The applied magnetic field during measurements was +5000 Oe.

3 Results and Discussions

3.1 Structural analysis

The study of crystal structure The X-ray diffraction (XRD) technique was used to examine Ni_{0.7}Zn_{0.3}Fe₂O₄ + 0.8 wt%V₂O₅ produced through ceramic process. X-ray diffraction is now utilized to characterise a wide range of materials, including single-crystal epitaxial thin films, polycrystal and powder combinations, and even randomly oriented amorphous materials. X-ray diffraction aids scientists in the development of new pharmaceuticals, the classification of rock formations based on their mineral components, and the understanding of how atom arrangement affects the behaviour of energy storage materials scientists, pushing their ability to engineer materials on an atomic level. In X-ray diffraction, the sample is put in the centre and irradiated by an X-ray beam that moves in a coordinated motion through the detector. The signals from the sample were plotted on a graph, with the peaks corresponding to the material's atomic structure. Typically, crystals are made up of a regular arrangement of atoms, with each atom consisting of a nucleus surrounded by an electron cloud. Because an X-wavelength ray's is similar to the atomic diameter in a lattice. Diffraction happens when x rays interact and is caused by the distance between the atoms in a crystal. It may be used to determine the distance between the atoms in a crystal. When the waves are aligned, the signal is amplified, resulting in constructive interference. When the waves are out of phase, the signal is lost, resulting in destructive interference. Elastic scattering took place, resulting in a repeating pattern of atoms or flakes separated by well-defined distances. The atomic planes scatter X-rays when they are exposed to them. Theta (θ) is the angle formed by the incident and dispersed particles. The scattered waves must be in phase, which means the second wave must travel the same number of wavelengths as the first. One-half of the wavelength is travelled on the incident side, while the other half is travelled on the scattered side in this scenario. There are no impurities in the identified strong peaks in the XRD patterns. All the values of Miller indices (hkl), Bragg's angle (2θ), $\sin \theta$, $\sin \theta/\lambda$ interplanar spacing (d), and intensity of various reflections of Ni_{0.7}Zn_{0.3}Fe₂O₄ + 0.8 wt%V₂O₅ are tabulated in Table 1.

The XRD peaks and the peak data have indicated the formation of the complete ferrite phase of the Ni_{0.7}Zn_{0.3}Fe₂O₄ + 0.8 wt%V₂O₅ sample. The presence of planes (220), (310), (311), (222), (400), (422), (511), and (440) in the XRD pattern confirmed the creation of single-phase; cubic geometry; spinel structure of the prepared Ni_{0.7}Zn_{0.3}Fe₂O₄ + 0.8 wt%V₂O₅ sample, as shown in Figure 1. Assuming that the vanadium oxide form melts around 670°C and forms a liquid phase. When a V₂O₅ additive is added to nickel-zinc ferrite, it improves the mass transport mechanism and generates a liquid phase during sintering. "The incorporation of V⁵⁺ ions into the spinel crystals of ferrite creates various com-

Table 1. Miller indices (hkl), Bragg's angle (2θ), $\sin \theta$, $\sin \theta/\lambda$, interplanar spacing (d), and intensity of various reflections of $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4 + 0.8 \text{ wt}\% \text{V}_2\text{O}_5$

(hkl)	2θ (deg)	θ (deg)	$\sin(\theta)$	$\sin \theta/\lambda$	d (Å)	I (a.u)	I/I_0
(220)	30.24	15.12	0.261	0.169	2.952	6025	55.088
(310)	33.24	16.62	0.286	0.186	2.692	4016	36.719
(311)	35.6	17.8	0.306	0.198	2.519	10937	100
(222)	37.26	18.63	0.319	0.207	2.411	3637	33.254
(400)	43.3	21.65	0.369	0.24	2.087	4550	41.602
(422)	53.68	26.84	0.452	0.293	1.706	4258	38.932
(511)	57.18	28.59	0.479	0.311	1.609	5662	51.769
(440)	62.78	31.39	0.520	0.338	1.479	6216	56.835

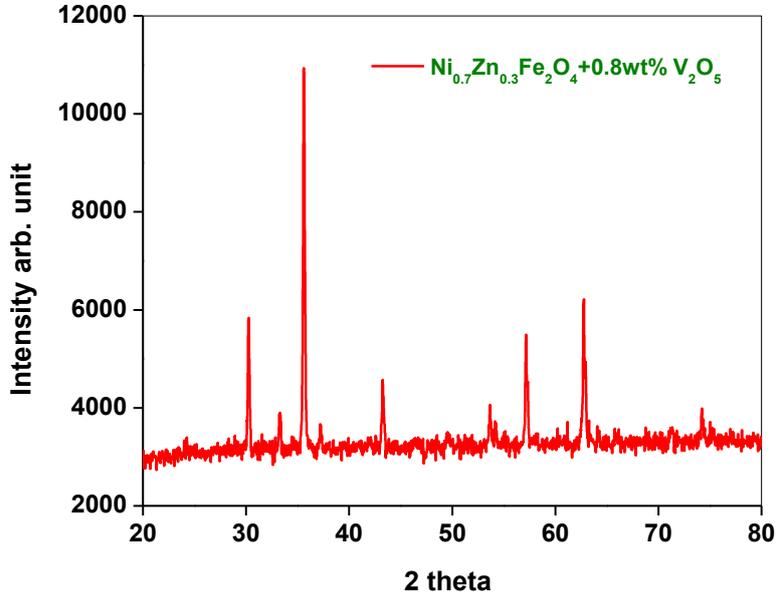


Figure 1. X-ray diffraction of $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4 + 0.8 \text{ wt}\% \text{V}_2\text{O}_5$.

Table 2. Hopping length L_A , L_B , tetrahedral bond d_{AX} octahedral bond d_{BX} tetra edge d_{AXE} , octa edge d_{BXE} and octa edge unshared d_{BXEu} of $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4 + 0.8 \text{ wt}\% \text{V}_2\text{O}_5$

L_A (Å)	L_B (Å)	d_{AX} (Å)	d_{BX} (Å)	d_{AXE} (Å)	d_{BXE} (Å)	d_{BXEu} (Å)
3.628	2.962	1.814	2.093	2.962	2.962	2.961

plicated structures in the system, such as the production of Fe²⁺ ions and the precipitation of Fe₂O₃ as the second phase of partial completion of the ferrite phase,” according to some literature. The values of Hopping length L_A , L_B , tetrahedral bond d_{AX} octahedral bond d_{BX} tetra edge d_{AXE} octa edge d_{BXE} , and octa edge unshared d_{BXEu} of Ni_{0.7}Zn_{0.3}Fe₂O₄ + 0.8 wt%V₂O₅ are given in Table 2.

The avg. lattice constants (a) for the Ni_{0.7}Zn_{0.3}Fe₂O₄ + 0.8 wt%V₂O₅ spinel ferrite was calculated using the following standard relation:

$$a = d_{hkl} \sqrt{h^2 + k^2 + l^2}, \quad (1)$$

where d_{hkl} the interplanar spacing of two planes, (a) is the lattice constant, and (hkl) is the Miller indices of each plane. The avg. lattice parameter (a) for Ni_{0.7}Zn_{0.3}Fe₂O₄ + 0.8 wt%V₂O₅ is $a = 8.380 \text{ \AA}$. According to the observations, the value of the lattice parameter over the V₂O₅ doping does not change to as much extend; and could be considered to be unchanged [32]. Also, V⁵⁺ ions have a smaller radius (0.59 Å) than that of the Fe³⁺ ions (0.64 Å) and Ni²⁺ ions (0.69 Å), the incorporation of V₂O₅ into lattice results in the creation of some Fe²⁺ ions which have an ionic radius (0.74 Å) [33]. The crystallite size of Ni_{0.7}Zn_{0.3}Fe₂O₄ + 0.8 wt%V₂O₅ was calculated by using the Debye-Scherrer method taking (311) plane of maximum intensity, which is mentioned by

$$t = \frac{k\lambda}{\beta \cos\theta}, \quad (2)$$

where λ is the X-ray wavelength, β is the full width at half maximum, θ is Bragg’s angle, and $k = 0.89$. An average crystallite size (t) obtained from XRD data for 0.8 wt%V₂O₅ doped Ni_{0.7}Zn_{0.3}Fe₂O₄ was found to be higher in comparison. The X-ray density (d_X) of Ni_{0.7}Zn_{0.3}Fe₂O₄ + 0.8 wt%V₂O₅ was calculated by the following relation:

$$d_X = 8M/(N_A^3), \quad (3)$$

where M is the molecular weight and N_A is the Avogadro’s number, α is lattice constant. The X-ray density for Ni_{0.7}Zn_{0.3}Fe₂O₄ + 0.8 wt%V₂O₅ was obtained as 5.303 g/cm³. The porosity was found to be 22.60% which was according to the expectations. The values of the structural parameters like lattice parameter α , X-ray density d_X , bulk density d_B , volume V , and porosity $P\%$ of Ni_{0.7}Zn_{0.3}Fe₂O₄ + 0.8 wt%V₂O₅ are depicted in Table 3.

Table 3. Lattice parameter α (Å), X-ray density d_X (g/cm³), bulk density d_B (g/cm³), volume V , and porosity P (%), and crystallite size of Ni_{0.7}Zn_{0.3}Fe₂O₄ + 0.8 wt%V₂O₅

α (Å)	d_X (g/cm ³)	d_B (g/cm ³)	V (Å ³)	P (%)	t
8.380	5.336	4.109	588.55	22.60	2.694

3.2 Infrared spectroscopy

The IR spectroscopy confirmed the functional group present and the bands associated with the $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4 + 0.8 \text{ wt}\% \text{V}_2\text{O}_5$. According to Waldron's research, the IR spectra of ferrite must be interpreted as continuously attached crystals [34]. As seen in Figure 2, the IR absorption bands of $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4 + 0.8 \text{ wt}\% \text{V}_2\text{O}_5$ are assigned to the vibration of metal ions (M-O) in the crystal lattice. The high-frequency band ν_1 (542 cm^{-1}) correspond to the intrinsic vibration of the tetrahedral group complexes $\text{Fe}^{3+}-\text{O}^{2-}$ and ν_2 (410.84 cm^{-1}) corresponds to the octahedral group complexes $\text{Fe}^{3+}-\text{O}^{2-}$.

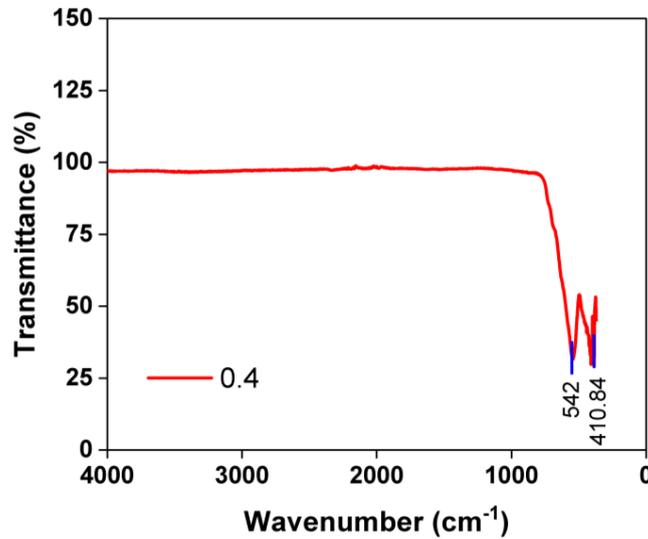


Figure 2. Infra-red spectra of $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4 + 0.8 \text{ wt}\% \text{V}_2\text{O}_5$.

3.3 DC-electrical resistivity

DC-electrical resistance was recorded by two probe methods in the temperature range $1000\text{--}2000 \text{ K}^{-1}$.

The DC-resistivity of $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4 + 0.8 \text{ wt}\% \text{V}_2\text{O}_5$ was calculated by standard formula

$$\rho = RA/t, \quad (4)$$

where R is the resistance, t is the thickness of the sample, and $A = \pi r^2$ is the area of the electrode in contact. The logarithm of DC-resistivity versus $1000/T$ (K^{-1}) temperature of $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4 + 0.8 \text{ wt}\% \text{V}_2\text{O}_5$ is shown in Figure 3 revealing the resistive property and marked a fall in resistivity with

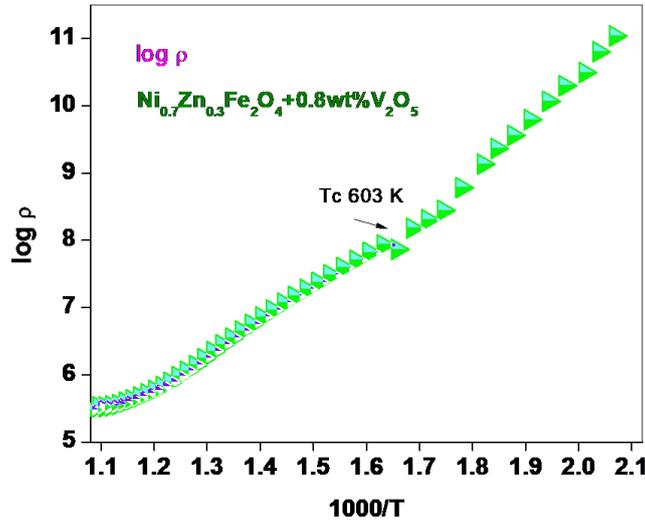


Figure 3. Temperature dependence DC-resistivity of Ni_{0.7}Zn_{0.3}Fe₂O₄ + 0.8 wt% V₂O₅.

hiking the temperature obeying the Arrhenius law [35]. Following the semiconducting characteristic of the material, increasing temperature causes the graph of DC-resistivity to decrease.

3.4 Magnetic property

The important magnetic parameters of Ni_{0.7}Zn_{0.3}Fe₂O₄ + 0.8 wt% V₂O₅ synthesized via ceramic method were determined using the pulse field hysteresis loop tracer technique on the device MAGNETA, at room temperature (see Figure 4). The applied magnetic field was kept at 5000 Oe. The values of saturation magnetization (M_s), remanence magnetization (M_r), and coercivity (H_c) were determined for Ni_{0.7}Zn_{0.3}Fe₂O₄ + 0.8 wt% V₂O₅ and Ni_{0.7}Zn_{0.3}Fe₂O₄ [23] are depicted in Table 4. It was found that the saturation magnetization (M_s) and remanence magnetization (M_r) over the addition of 0.8 wt% V₂O₅ were recorded

Table 4. Saturation magnetization (M_s), remanence magnetization (M_r), and coercivity (H_c), magnetic remanence to saturation ratio (M_r/M_s), and magneton number for Ni_{0.7}Zn_{0.3}Fe₂O₄ + 0.8 wt% V₂O₅

	M_s (emu/gm)	M_r (emu/gm)	H_c (Oe)	M_r/M_s	n_B (μ_b)
Ni _{0.7} Zn _{0.3} Fe ₂ O ₄ + 0.8 wt% V ₂ O ₅	79.11	2.44	75.24	33.84	2.576
Ni _{0.7} Zn _{0.3} Fe ₂ O ₄ [23]	74.60	2.26	18.71	33.00	2.429

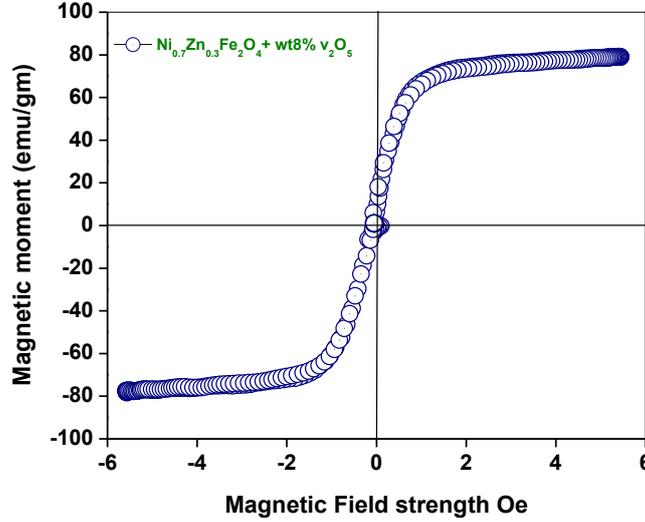


Figure 4. $M-H$ curve of magnetization for $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4 + 0.8 \text{ wt}\% \text{V}_2\text{O}_5$.

to be increased in comparison to the $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4$ [23]. A drastic change in the coercivity (H_c) value was seen as an effect of 0.8 wt% V_2O_5 .

4 Conclusions

The ceramic approach was used to make $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4 + 0.8 \text{ wt}\% \text{V}_2\text{O}_5$ in this study. The standard formula was used to determine the structural values acquired from X-ray diffraction data. The inverse type, single-phase, cubic spinel ferrite structure of $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4 + 0.8 \text{ wt}\% \text{V}_2\text{O}_5$ produced using ceramic technique has been validated. The lattice constant was reported as 8.380 Å, which is consistent with the literature and was found to be unaffected by the addition of 0.8 wt% V_2O_5 . The X-ray density (d_X) of $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4 + 0.8 \text{ wt}\% \text{V}_2\text{O}_5$ was obtained to be 5.336 g/cm³. Hopping length L_A , L_B , tetrahedral bond d_{AX} , octahedral bond d_{BX} , tetra edge d_{AXE} , octa edge d_{BXE} , and octa edge unshared d_{BXEu} of $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4 + 0.8 \text{ wt}\% \text{V}_2\text{O}_5$ were derived from the XRD data which are following the reported range in literature. The infrared spectroscopy (IR) has confirmed the ferrite phase formation in $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4 + 0.8 \text{ wt}\% \text{V}_2\text{O}_5$. The logarithmic plot of DC-resistivity has suggested the semiconducting behavior of $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4 + 0.8 \text{ wt}\% \text{V}_2\text{O}_5$. Also, an increase in saturation magnetization (M_s), remanence magnetization (M_r), and magneton number (n_B) of $\text{Ni}_{0.7}\text{Zn}_{0.3}\text{Fe}_2\text{O}_4 + 0.8 \text{ wt}\% \text{V}_2\text{O}_5$ have been recorded which may suit several application areas.

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